

**SYNTHESIS AND CHARACTERIZATION OF
LEAD ZIRCONATE TITANATE
(Pb[Zr_{0.52}Ti_{0.48}]O₃) PROPERTIES VIA HIGH
ENERGY PLANETARY BALL MILLING**

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DOCTOR OF PHILOSOPHY

UNIVERSITI MALAYSIA PAHANG

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(Pb[Zr_{0.52}Ti_{0.48}]O₃) PROPERTIES VIA HIGH ENERGY PLANETARY BALL
MILLING**

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Thesis submitted in fulfillment of the requirements
for the award of the degree of
Doctor of Philosophy in Advance Ceramic Material

Faculty of Manufacturing Engineering
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TABLE OF CONTENTS

	Page
THESIS CONFIDENTIAL STATUS	i
SUPERVISOR'S DECLARATION	ii
STUDENT'S DECLARATION	iii
ACKNOWLEDGEMENT	iv
ABSTRACT	v
ABSTRAK	vi
TABLE OF CONTENTS	vii
LIST OF TABLES	xi
LIST OF FIGURES	xii
LIST OF SYMBOLS	xvii
LIST OF ABBREVIATIONS	xx

CHAPTER 1 INTRODUCTION

1.1	Ceramic Materials	1
1.2	Problem Statement	3
1.3	Research Objectives	4
1.4	Research Scope	4
1.5	Thesis Organization	5

CHAPTER 2 LITERATURE REVIEW

2.1	Introduction	7
2.2	Perovskite Structure	8
2.3	Lead Zirconate Titanate (PZT)	9
2.3.1	Phase Diagram for PZT Binary System	11
2.4	Polarization in PZT Ceramics	12
2.4.1	Piezoelectricity	14
2.5	Dielectric and Piezoelectric Properties of PZT Ceramic	16
2.5.1	Dielectric Material	16
2.5.2	Dielectric Properties	16
2.5.3	Polarization of Dielectric	19
2.6	Piezoelectric Theory	21

2.6.1	Piezoelectric Properties	23
2.6.2	Mechanical Quality Factor	24
2.6.3	Piezoelectric Coupling Coefficient	24
2.6.4	Piezoelectric Charge Coefficient	26
2.6.5	Young Modulus Factor	26
2.6.6	Piezoelectric Voltage Constant	27
2.6.7	Parameters for Piezoelectric Material	27
2.7	Microstructure Effect	29
2.8	Processing Effect	32
2.9	Electrode/ Sample Contact Effect	35
2.10	Summary	36

CHAPTER 3 MATERIALS AND METHOD

3.1	Introduction	38
3.2	Starting Materials	42
3.3	Processing Steps	43
3.3.1	Composition Formulation	43
3.3.2	Milling Process	43
3.3.3	De-agglomeration	44
3.3.4	Uniaxial Pressing	44
3.3.5	Sintering	46
3.3.6	Sample Contact/Electrode	47
3.4	Sample Preparation and Characterization Techniques	47
3.4.1	X-ray Diffraction (XRD)	47
3.4.2	Scanning Electron Microscope (SEM)	48
3.4.3	Density Test	49
3.4.4	Thermal Analysis	49
3.4.5	Poling Process	50
3.4.6	Dielectric Properties Measurement	50
3.4.7	Piezoelectric Properties Measurement	53
3.5	Summary	54

CHAPTER 4 RESULTS AND DISCUSSION

4.1	Introduction	55
4.2	Characterization of Starting Materials	55
4.2.1	Plumbum Oxide (PbO) Powder	56
4.2.2	Zirconium Oxide (ZrO ₂) Powder	58
4.2.3	Titanium Oxide (TiO ₂) Powder	60
4.3	Synthesis of PZT	62
4.3.1	Phase Analysis of Milled Powders	63
4.3.2	Morphology of Milled Powders	64

4.3.3	Thermogravimetry / Differential Thermal Analysis	68
4.4	Sintering Process	70
4.4.1	SINTERING I	70
4.4.1.1	Phase Analysis	71
4.4.1.2	Microstructure of Sintered PZT Ceramics	73
4.4.1.3	Density Measurement	75
4.4.1.4	Dielectric Measurement	76
4.4.1.5	Piezoelectric Property Measurements	79
4.4.2	PART I	82
4.4.2.1	XRD Analysis	82
4.4.2.2	Surface Microstructure	85
4.4.2.3	Fracture Surface Microstructure	87
4.4.2.4	Density Measurement	89
4.4.2.5	Dielectric Measurement	90
4.4.2.6	Measurement of Piezoelectric Properties	94
4.4.3	PART II	97
4.4.3.1	Phase Analysis	98
4.4.3.2	Surface Microstructure	100
4.4.3.3	Fracture Surface Microstructure	103
4.4.3.4	Density Measurement	105
4.4.3.5	Dielectric Measurement	106
4.4.3.6	Piezoelectric Property Measurements	109
4.4.4	PART III	112
4.4.4.1	Phase Analysis	112
4.4.4.2	Surface Microstructure	115
4.4.4.3	Fracture Surface Microstructure	116
4.4.4.4	Density Measurement	119
4.4.4.5	Dielectric Measurement	121
4.4.4.6	Measurement of Piezoelectric Properties	124
4.4.5	Summary	128

CHAPTER 5 CONCLUSION AND RECOMMENDATION

5.1	Conclusion	129
5.2	Recommendation for Future Research	130

REFERENCES 132

APPENDICES

A	Weight of 30 g of Starting Materials	148
B	Calculation on Starting Materials Needed to Prepare a Batch of 30 g of Pb(Zr _{0.52} Ti _{0.48})O ₃	149

C	Powder Diffraction File of PbO	150
D	Powder Diffraction File of ZrO ₂	152
E	Powder Diffraction File of TiO ₂	154
F	Powder diffraction file of Pb(Zr _{0.52} Ti _{0.48})O ₃	156
G	Grain Size Distribution	158
H	Relative Density and Apparent Porosity	166
I	Piezoelectric coupling factor (k_p , k_t) and Mechanical Quality factor (Q_m)	168
LIST OF PUBLICATIONS AND CONFERENCES		170

LIST OF TABLES

Table No.	Title	Page
2.1	The value of dielectric constant for some materials at room temperatures	19
2.2	Basic terminology of piezoelectric material	22
2.3	Standard piezoelectric properties of piezoelectric material	28
3.1	Comparison of PZT ceramic materials synthesis routes	42
3.2	Raw materials, suppliers and purity of the powders	42
3.3	Weight of starting materials to prepare 30 g batch powder of pure PZT	43
4.1	Lattice parameter measurement of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ with various milling durations according to peak (002), (200) and (112), (121).	72
4.2	Mean grain size and standard deviation of PZT samples sintered at 1000 °C for 1 hour soaking time.	75
4.3	Dielectric parameters of the PZT ceramics measured at 1 MHz sintered at 1000 °C for 1 hour.	79
4.4	Lattice parameter measurement of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered pellets at different sintering temperature for 1 hour soaking time.	84
4.5	Mean grain size and standard deviation of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered at different temperature for 1 hour duration.	87
4.6	Lattice parameter measurement of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered pellets at different sintering temperature for 2 hours soaking time.	100
4.7	Mean grain size and standard deviation of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered at different temperature for 2 hours duration.	102
4.8	Lattice parameter measurement of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered pellets at different sintering temperature for 3 hours soaking time.	114
4.9	Mean grain size and standard deviation of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered at different temperature for 3 hours duration.	117

LIST OF FIGURES

Figure No.	Title	Page
2.1	Schematic of (a) direct and (b) converse piezoelectric effect	8
2.2	Perovskite simple cubic structures	9
2.3	Perovskite crystal structure of PZT. Unit cell are in the (a) cubic, (b) tetragonal, (c) orthorhombic and (d) rhombohedral	10
2.4	Phase diagram system of the PZT binary solid solution	11
2.5	Schematic illustration of the poling process	14
2.6	A typical polarization-voltage hysteresis loop (a) and domain in ferroelectrics movement (b)ceramic material	15
2.7	A parallel-plate capacitor (a) when a vacuum is present and (b) when a dielectric material is present	18
2.8	Schematic representations of different mechanisms of polarization	20
2.9	The relative permittivity (dielectric constant) as function of frequency	21
2.10	Limiting grain size for ferroelectric domain formation	29
3.1	Process flow of experimental work.	41
3.2	Basic flow chart of the mixed route for synthesis of PZT ceramics.	41
3.3	(a) Retch high energy planetary ball milling, (b) Tungsten carbide vial for dry milling process, and (b) Tungsten carbide balls as media.	44
3.4	A die mold set for uniaxial pressing.	45
3.5	Temperature profile for sintering of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ ceramics.	46
3.6	Equipments for dielectric material measurement.	51
3.7	Basic flow chart for dielectric material measurements at high frequency.	52
3.8	A d_{33} meter with attached apparatus for holding specimen.	53
3.9	An impedance (Z) versus frequency trace extracted from the Impedance Analyzer	54
4.1	SEM micrograph of PbO powders	56
4.2	EDX analysis of composition of PbO powder.	57

4.3	XRD pattern of PbO powder	57
4.4	SEM micrograph of ZrO ₂ powder	58
4.5	EDX analysis of composition of ZrO ₂ powder.	59
4.6	XRD pattern of ZrO ₂ powder	59
4.7	SEM micrograph of TiO ₂ powder	60
4.8	EDX analysis of composition of TiO ₂ powder.	61
4.9	XRD pattern of TiO ₂ powder	61
4.10	Milled powder of PbO-ZrO ₂ -TiO ₂ .	62
4.11	XRD patterns for milled powder of PbO, ZrO ₂ and TiO ₂ .	63
4.12	SEM image and EDX spectra of milled PZT powders milling for (i) 20, (ii) 40, and (iii) 60 hours.	68
4.13	TGA-DTA results for 40 hours milled PZT powder	69
4.14	XRD analysis for PZT samples sintered at 1000 °C for 1 hour soaking time with different milling duration.	71
4.15	Close-up of X-ray diffraction patterns of Pb(Zr _{0.52} Ti _{0.48})O ₃ sintered pellets with different milling durations	72
4.16	SEM micrographs of surface and fracture surface for PZT samples sintered at 1000 °C for 1 hour sintering time with different milling duration: (a) and (b) 20 hours, (c) and (d) 40 hours, (f) and (g) 60 hours.	74
4.17	Grain size distribution of PZT samples sintered at 1000 °C for 1 hour soaking time with different milling durations.	74
4.18	Relative density and apparent porosity of PZT sample sintered at 1000 °C for 1 hour soaking time with different milling durations.	75
4.19	Dielectric constant at high frequency (1 MHz – 1 GHz) for pure PZT sintered at 1000 °C for 1 hour with different milling durations.	76
4.20	Dielectric loss at high frequency for pure PZT sintered at 1000 °C for 1 hour with various milling durations. Inset is close up of dielectric loss value from 1 MHz to 100 MHz.	78
4.21	The piezoelectric charge coefficient (d_{33}) for pure PZT samples sintered at 1000 °C for 1 hour soaking time with various milling durations.	80
4.22	The piezoelectric electromechanical coupling factor (k_v, k_p) for pure PZT samples sintered at 1000 °C for 1 hour soaking time with various milling durations.	81

4.23	The mechanical quality factor (Q_m) for pure PZT samples sintered at 1000 °C for 1 hour soaking time with various milling durations.	82
4.24	XRD analysis for PZT samples sintered at various temperatures for 1 hour soaking time.	84
4.25	Close-up of X-ray diffraction patterns of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered pellets at different sintering temperature for 1 hour soaking time.	84
4.26	SEM micrographs of PZT samples sintered at different sintering temperature for 1 hour sintering time.	86
4.27	Grain size distribution of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered at different temperature for 1 hour duration.	87
4.28	SEM images of fracture surface sintered at different temperature or 1 hour soaking time.	88
4.29	EDX analysis of fracture surface morphology for sample sintered at 1100 °C for 1 hour soaking time.	89
4.30	Variation in relative density and porosity of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ ceramics as a function of different sintering temperature for 1 hour soaking time.	90
4.31	Frequency dependence of the dielectric constant for PZT pellets sintered at various sintering temperature for 1 hour soaking time.	91
4.32	Frequency dependance of the dielectric loss for PZT pellets sintered at different sintering temperature for 1 hour soaking time. Inset is close up of dielectric loss at frequency 1-100 MHz.	94
4.33	The piezoelectric charge coefficient (d_{33}) for pure PZT samples sintered at various sintering temperatures for 1 hour soaking time.	95
4.34	The piezoelectric coupling factor (k_t and k_p) for pure PZT samples sintered at various sintering temperatures for 1 hour soaking time.	96
4.35	The mechanical quality factor (Q_m) for pure PZT samples sintered at various sintering temperatures for 1 hour soaking time.	97
4.36	XRD analysis of PZT samples sintered at different sintering temperature with 2 hours soaking time.	98
4.37	Close-up of X-ray diffraction patterns of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered pellets at different sintering temperature for 2 hours soaking time.	99
4.38	SEM micrographs of PZT samples sintered at different sintering temperature for 2 hours sintering times.	101
4.39	Grain size distribution of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered at differenttemperature for 2 hours duration.	102
4.40	SEM images of fracture surface sintered at different temperature for 2 hours soaking time.	104

4.41	EDX analysis of fracture surface morphology for sample sintered at 1100 °C for 2 hours soaking time.	105
4.42	Variation in relative density and porosity of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ ceramics as a function of different sintering temperature for 2 hours soaking time.	106
4.43	Frequency dependence of the dielectric constant for PZT pellets sintered at various sintering temperature for 2 hours soaking time.	107
4.44	Frequency dependence of the dielectric loss for PZT pellets sintered at different sintering temperature for 2 hours soaking time. Inset is close up of dielectric loss at frequency 1 – 100 MHz.	109
4.45	The piezoelectric charge coefficient (d_{33}) for pure PZT samples interred at various sintering temperatures for 2 hours soaking time.	110
4.46	The piezoelectric coupling factor (k_p, k_t) for pure PZT samples sintered at various sintering temperatures for 2 hours soaking time.	111
4.47	The mechanical quality factor (Q_m) for pure PZT samples sintered at various sintering temperatures for 2 hours soaking time.	112
4.48	XRD analysis of PZT samples sintered at different sintering temperature with 3 hours soaking time.	113
4.49	Close-up of X-ray diffraction patterns of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered pellets at different sintering temperaturefor 3 hours soaking time.	114
4.50	SEM micrographs of PZT samples sintered at different sintering temperature for 3 hours sintering times.	116
4.51	Grain size distribution of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ sintered at different temperature for 3 hours duration.	117
4.52	SEM images of fracture surface sintered at different temperatures for 3 hours sintering duration	118
4.53	EDX analysis of fracture surface morphology for sample sintered at 1100 °C for 3 hours soaking time.	119
4.54	Variation in relative density and apparent porosity of $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ ceramics as a function of different sintering temperature for 3 hours soaking time.	120
4.55	Frequency dependence of the dielectric constant for PZT pellets sintered at various sintering temperature for 3 hours soaking time.	122
4.56	Frequency dependence of the dielectric loss for PZT pellets sintered at different sintering temperature for 3 hours soaking time. Inset is close up of dielectric loss at frequency 1 – 100 MHz	123
4.57	The piezoelectric charge coefficient (d_{33}) for pure PZT samples sintered at various sintering temperatures for 3 hours soaking time.	125

4.58	The piezoelectric coupling factor (k_p, k_t) for pure PZT samples sintered at various sintering temperatures for 3 hours soaking time.	126
4.59	The mechanical quality factor (Q_m) for pure PZT samples sintered at various sintering temperatures for 3 hours soaking time.	127

LIST OF SYMBOLS

%	Percentage
°	Degree
θ	Theta
°C	Degree Celsius
Å	Armstrong
M _D	Dry Weight
M _S	Suspended Weight
M _W	Saturated Weight
MPa	Mega Pascal
MHz	Mega Hertz
GHz	Giga Hertz
kHz	Kilo Hertz
ml	Milliliter
g	Gram
rpm	Rotation per Minutes
pC/N	Coulomb per Newton
nm	Nanometer
μm	Micrometer
mm	Millimeter
cm	Centimeter
kV	Kilo Volts
K	Kelvin
m	Meter
wt %	Weight percent

Ω	Ohm
s	Second
min	Minutes
ϵ	Permittivity
$\tan \delta$	Tangent lost or dissipation factor
C	Capacitance
V	Volt
T_c	Curie temperature
d_{33}, d_{31}, d_{15}	Piezoelectric Charge Constants
$s_{33}^D, s_{11}^D, s_{11}^E,$ $s_{33}^E, s_{55}^E, s_{55}^D$	Elastic Constants
g_{33}, g_{31}	Piezoelectric Voltage Constants
Q_m	Mechanical Quality Factor
k_{33}, k_{31}	Piezoelectric Coupling Factors
k_p	Planar Coupling Factor
k_t	Thickness Coupling Factor
f_a	Anti-Resonance Frequency
f_r	Resonance Frequency
Z_r	Impedance at Resonance Frequency
P_s	Spontaneous Polarization
P_r	Remanent Polarization
F_R	Rhombohedral Phase
F_{RL}	Low Temperature Rhombohedral Phase
F_{RH}	High Temperature Rhombohedral Phase
F_T	Ferroelectric Tetragonal

P Polarization
E Electric Field

LIST OF ABBREVIATIONS

Pb(Zr _{0.52} Ti _{0.48})O ₃ (PZT)	Lead Zirconate Titanate
PbO	Plumbum Oxide
ZrO ₂	Zirconia Oxide
TiO ₂	Titanium Oxide
ABO ₃	Perovskite Family
MgTiO ₃	Magnesium Titanate
PbTiO ₃	Plumbum Titanate
CaTiO ₃	Calcium Titanate
BaTiO ₃	Barium Titanate
Pt	Platinum
Ag	Silver
RuO _x	Ruthenium Oxides
MPB	Morphotropic Phase Boundary
SEM	Scanning Electron Microscopy
XRD	X-ray Diffraction
EDX	Energy-dispersive X-ray Spectroscopy
EDS	Energy Dispersive Spectrometry
DTA	Differential Thermal Analysis
TGA	Thermal Gravimetry Analysis

ABSTRACT

Pb(Zr_{0.52}Ti_{0.48})O₃ piezoelectric was successfully prepared by solid state reaction method via high energy planetary ball milling. This method which implements a single firing step where the calcinations process has been skipped thus successfully controls the higher evaporation amount of PbO during sintering process. Optimization is important on sintering parameters when using a single firing process. With the optimum processing parameters and sintering conditions (temperature and duration), the excellent PZT formation and optimum electrical properties of PZT ceramics was produce. The properties of piezoelectric materials were modified by different milling duration (20, 40 and 60 hours) and sintering parameters (temperature: 1000, 1050, 1100, 1150 and 1200 °C; soaking time: 1, 2 and 3 hours). Based on the study, sample milled for 40 hours show suitable milling duration with SEM analysis promotes fine particles and densification, thus achieved an excellent dielectric and piezoelectric properties. Preliminary works have been done in order to find suitable sintering temperature and durations to obtain the best PZT formation with good piezoelectric properties. This study continued to the next phase with 40 hours and sintered at different sintering temperatures (1050, 1100, 1150 and 1200 °C) and durations (1, 2 and 3 hours). According to the results, XRD analysis indicate that samples sintered at different sintering temperature and duration show single perovskite phase and no crystal structure changes were detected. However, sample sintered at 1200 °C for longer sintering duration (3 hours), new phase which mainly of excess PbO oxides were identified. SEM analysis was carried out and shows that, fine and dense grain sizes were achieved after sintering at 1050, 1100 °C for 1 hour. Increasing the sintering temperature and duration, larger and irregular grain size were observed. SEM observation indicated that sample sintered at 1200 °C for 2 hours durations shows better grain morphology whereby homogenous and dense microstructure were obtained. Increasing of the sintering temperature and durations, produces an increase of the grain size, increase in dielectric and piezoelectric properties thus decrease the value of dielectric loss for PZT ceramic. However, higher sintering temperature (1200 °C) and longer sintering durations (3 hours), the relative density slightly decreases to about 92 % of the theoretical density and causes the electrical properties to decrease. This phenomenon happens due to the evaporation of PbO whereby the existence of PbO phase was detected in the XRD patterns. Based on the results, sample sintered at 1200 °C for 2 hours show optimum sintering parameter and exhibits excellent room temperature electrical properties, which are listed as follows: $\epsilon_r = 2400$, $\tan \delta = 0.009$, $d_{33} = 256$ pC/N, $k_t = 0.48$, $k_p = 0.42$ and $Q_m = 1716$.

ABSTRAK

Piezoelektrik $\text{Pb}(\text{Zr}_{0.52}\text{Ti}_{0.48})\text{O}_3$ telah berjaya dihasilkan menggunakan kaedah tindak balas pepejal melalui proses pengisar bebola planetari. Kaedah ini menggunakan satu proses pembakaran yang mana pengkalsinan dapat dilakukan dan sekali gus Berjaya mengawal pemeruapan jumlah PbO semasa proses pensinteran. Pengoptimunan adalah penting untuk parameter pensinteran apabila menggunakan satu proses pembakaran. Dengan optimum parameter pemprosesan dan keadaan pensinteran yang bagus (suhu dan tempoh), pembentukan PZT yang baik dan sifat elektrik yang optimum dapat dihasilkan. Sifat – sifat bahan seramik piezoelektrik diubah suai dengan tempoh pengisaran (20, 40 dan 60 jam) dan parameter pensinteran yang berbeza (suhu pensinteran: 1000, 1050, 1100, 1150 dan 1200 °C; tempoh pensinteran: 1, 2 dan 3 jam). Hasil daripada kajian, sampel yang dikisar selama 40 jam menunjukkan tempoh pengisaran paling sesuai dengan analisis mikrostruktur menunjukkan saiz partikel yang seragam dengan ketumpatan yang tinggi dan memperolehi nilai lesapan dielektrik dan sifat piezoelektrik yang bagus. Penyelidikan awal telah dijalankan untuk mengenalpasti suhu dan tempoh pensinteran yang sesuai bagi memperoleh pembentukan PZT dan sifat piezoelektrik yang bagus. Kajian ini diteruskan kepada fasa lain dengan menggunakan tempoh pengisaran 40 jam dan sampel disinter pada suhu (1050, 1100, 1150 dan 1200 °C) dan masa (1, 2 dan 3 jam) pensinteran yang berbeza. Hasil daripada keputusan ujikaji, analisis belauan sinar-X menunjukkan sampel yang disinter dengan suhu dan tempoh pensinteran yang berbeza menghasilkan satu fasa tunggal (PZT) dan tiada perubahan struktur kristal. Walaubagaimanapun, sampel yang disinter pada suhu 1200 °C untuk tempoh pensinteran yang lama (3 jam), fasa baru iaitu lebihan oksida PbO telah dikenal pasti. Analisis mikrostruktur menunjukkan, butiran saiz yang kecil dan sekata telah diperolehi selepas pensinteran pada suhu 1050 dan 1100 °C selama 1 jam. Apabila suhu dan masa pensinteran ditingkatkan, saiz butiran yang membesar dan tidak sekata terhasil. Pemerhatian pada mikrostruktur menunjukkan sampel yang disinter pada suhu 1200 °C selama 2 jam memaparkan morfologi butiran yang lebih baik yang mana butiran lebih sekata dan seragam yang jelas kelihatan. Kenaikan suhu dan tempoh pensinteran menghasilkan saiz butir yang besar, peningkatan pada sifat dielektrik, menurunkan nilai lesapan dan meningkatkan nilai sifat piezoelektrik bagi seramik PZT. Namun, lebih tinggi suhu pensinteran menjangkau 1200 °C untuk tempoh pensinteran yang panjang (3 jam), nilai ketumpatan menurun sehingga 92 % daripada ketumpatan teori dan sifat elektrik seramik PZT menurun. Hal ini berlaku disebabkan pemeruapan PbO yang mana fasa PbO telah dikenal pasti melalui analisis belauan sinar X-ray. Dalam kajian ini, sampel yang disinter pada suhu 1200 °C selama 2 jam menunjukkan parameter pensinteran yang paling sesuai dan mencapai sifat elektrik yang paling bagus seperti yang disenaraikan: $\epsilon_r = 2400$, $\tan \delta = 0.009$, $d_{33} = 256 \text{ pC/N}$, $k_t = 0.48$, $k_p = 0.42$ and $Q_m = 1716$.

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